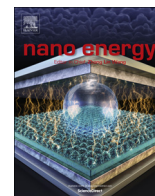




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# Bunchy TiO<sub>2</sub> hierarchical spheres with fast electron transport and large specific surface area for highly efficient dye-sensitized solar cells



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## ABSTRACT

The exploration of TiO<sub>2</sub> nanostructures with fast electron transport and large specific surface area is essential for their applications in solar cells. For this aim, bunchy TiO<sub>2</sub> hierarchical spheres (BS-TiO<sub>2</sub>), consisting of multiple TiO<sub>2</sub> microspheres constructed by nanosheets and the 1D TiO<sub>2</sub> nanobelt, were rationally designed and fabricated. BS-TiO<sub>2</sub> enables the much fast charge transport properties in the photoanode (with a long electron diffusion length of 36 μm) and shows large specific surface area (higher than 200 m<sup>2</sup>/g), leading to excellent electron collection and light harvesting, which enable the increased conversion efficiency of the dye-sensitized solar cell by 34% for BS-TiO<sub>2</sub> photoanode (7.5%) compared to nanoparticle (P25) photoanode. This work provides an important approach for the rational design of the TiO<sub>2</sub> nanostructures, which also implies the application of the bunchy TiO<sub>2</sub> hierarchical spheres in other type of solar cells, artificial photosynthesis, and other optoelectronic applications.

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## 1. Introduction

Dye-sensitized solar cells (DSCs), with the advantages of high efficiency, low-cost and relatively high stability [1,2], are a photoelectrochemical system with dye-adsorbed porous-structured oxide (typically TiO<sub>2</sub>) as the photoanode. As the key component of a DSC, the photoanode governs the electron collection efficiency and light harvesting efficiency, and thus the overall cell conversion efficiency. The light harvesting efficiency, depending on the surface area of electrode materials for dye adsorption, is normally associated with the high porosity and small size of building blocks (e.g., nanoparticles). However, high surface area films consisting of nanocrystalline particles yield high grain boundary densities, which increase the charge carrier recombination through loss of carrier mobility [3]. The rapid charge transport property usually relies on the straight conducting pathways [3–5], such as nanotubes, nanorods, while the dye-adsorption is limited in these films due to the small surface area of these films. Though the trade-off between the light harvesting and the carrier transport is an inevitable and universal problem in DSCs, the solutions have not been well explored.

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One promising strategy to solve the problem is to integrate the small sized building blocks into the micro sized architectures. The architectures, such as nanocrystallite aggregates, dendritic spheres, typically exhibit three-dimensional (3D) spherical morphology [6–10]. These TiO<sub>2</sub> architectures can provide large surface areas for dye-loading favored by the small building blocks, and improved electron collection by the reduced defects or grain boundaries inner the spherical architecture. In addition, it has been found that the micro and mesoporous TiO<sub>2</sub> materials also allow the light waves to penetrate deep inside the photoanode and make the photoanode more efficient in light harvesting [11–13]. The DSCs using the photoanodes fabricated from these architectures typically show increased light harvesting compared with traditional P25 or small sized TiO<sub>2</sub> nanoparticles [14]. However, the increase in the cell efficiency is limited or is even lower with the introduction of 3D spherical architecture [15], probably due to the small size of the inner grains in the micro-architectures which is not able to form the built-in potential and increases the recombination with their large surface area.

Substantial research efforts for overcoming the competition between the light harvesting and charge collection have been recently focused on building nanoblocks on the one-dimensional (1D) trunks (e.g., nanorods, nanotubes), i.e., hierarchical architectures [4,5,16]. The electron transport is expected to via the 1D trunks, while the surface nanostructures can increase the surface area for dye-loading [5,16]. In addition, the light scattering can

also be enhanced in the hierarchical architectures. However, due to the small surface area of 1D trunks, limited nanoblocks can be built on their surface, which limits the surface area of the hierarchical architecture for dye-loading. For example, with TiO<sub>2</sub> nanosheets deposited on SnO<sub>2</sub> nanotubes, the specific surface area is only increased by a small extent from 70 m<sup>2</sup>/g to no more than 85 m<sup>2</sup>/g [5]. Therefore, though the DSCs using these kinds of hierarchical architectures as photoanode materials show higher conversion efficiencies than the cells using their 1D trunk counterparts, there are still plenty room to enhance the cell efficiencies.

Herein, we describe the fabrication of the bunchy TiO<sub>2</sub> hierarchical structure consisting of 3D spheres stringed by the primary 1D nanobelt trunk. The modified fabrication processes enable the deposition of 3D integrated microspheres constructed by nanosheets, instead of individual nanosheets, on the 1D nanobelt trunk, which allows the novel bunchy TiO<sub>2</sub> spheres (BS-TiO<sub>2</sub>) possessing the advantages both of the 3D architectures and the hierarchical architectures. It is demonstrated that the BS-TiO<sub>2</sub> based photoanode performs much better than the standard TiO<sub>2</sub> nanocrystalline (P25-TiO<sub>2</sub>) based photoanode and the P25-TiO<sub>2</sub>/scattering bilayer photoanode in the DSCs. In light of these results, it is concluded that photoanodes composed of BS-TiO<sub>2</sub> are attractive for solar cells, artificial photosynthesis, and other optoelectronic applications.

## 2. Results and discussion

To achieve fast electron transport, 1D nanobelt trunks with direct path for electron transport is selected. To get a large surface area for dye-loading and provide the light-scattering paths for light trapping inside the photoanode, it is rational to construct the micro-sized 3D integrated spherical structures with nanoscale building blocks. Hence, the architecture with spherical TiO<sub>2</sub> grown on nanobelt will increase effectively the light absorption and fast transport the photogenerated electrons collected by the nanoscale building blocks, leading to the high light harvesting efficiency and electron collection efficiency. A key problem in fabricating such TiO<sub>2</sub> structures is how to grow the integrated 3D spheres, other than nanoscale building blocks themselves, on the TiO<sub>2</sub> nanobelts. Hence, a modified template-free hydrothermal procedure with Ag cations as the agent and the TiO<sub>2</sub> nanobelts containing TiO<sub>2</sub>-B phase was used for fabrication of such TiO<sub>2</sub> structures, the schematic processes of which are depicted in Fig. 1. The tetrabutyl

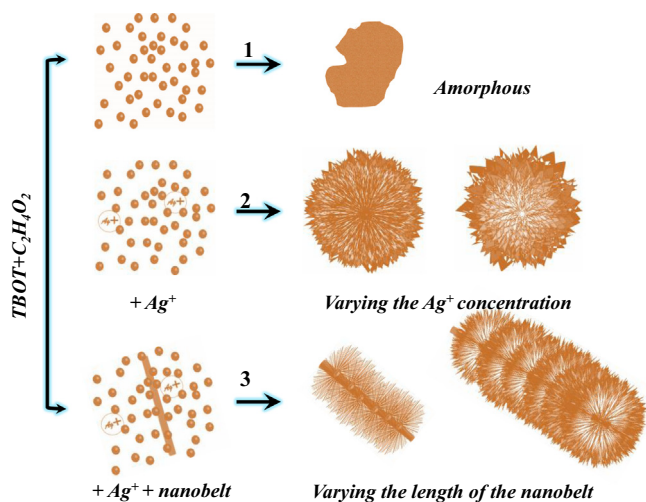
titanate (TBOT) solutions with acetic acid (C<sub>2</sub>H<sub>4</sub>O<sub>2</sub>) as the solvent were used in the reaction. In the typical procedure with TBOT and C<sub>2</sub>H<sub>4</sub>O<sub>2</sub> as the reactants (scheme 1), the product is typically amorphous. With the assistance of Ag cations and the fine adjusting of the content of Ag cations in the reaction solution (scheme 2), individual 3D spherical TiO<sub>2</sub> (S-TiO<sub>2</sub>) constructed by ultra-thin nanosheets can be fabricated. In condition of the TiO<sub>2</sub> nanobelts with TiO<sub>2</sub>-B as the surface crystalline phase presented in the reaction solution (scheme 3), the 3D spheres grow on the surface of the nanobelt, forming the bunchy spherical TiO<sub>2</sub> hierarchical structures (BS-TiO<sub>2</sub>). It is worth noting that TiO<sub>2</sub>-B phase in TiO<sub>2</sub> nanobelt is crucial for the growth of the 3D spheres on its surface, as TiO<sub>2</sub> nanobelt with pure anatase phase is not able to bunch the TiO<sub>2</sub> spheres.

Fig. 2 shows the typical SEM images of the products. The as-prepared S-TiO<sub>2</sub> shows the spherical morphology consisting of a large number of thin TiO<sub>2</sub> nanosheets; the outer diameter was approximately 2–4 μm (as shown in Fig. 2a–c). The density of the nanosheets on S-TiO<sub>2</sub> can be adjusted by varying the concentration and/or the ratio of Ag cations and the precursors. In the presence of TiO<sub>2</sub> nanobelt in the reaction solution, TiO<sub>2</sub> spheres integrated by nanosheets grow directly on the nanobelt, forming the bunchy spherical morphology. The outer diameter of BS-TiO<sub>2</sub> is approximate 1–2 μm, while its length is approximate 1–10 μm according to the length of inner TiO<sub>2</sub> nanobelt (Fig. 2d–f).

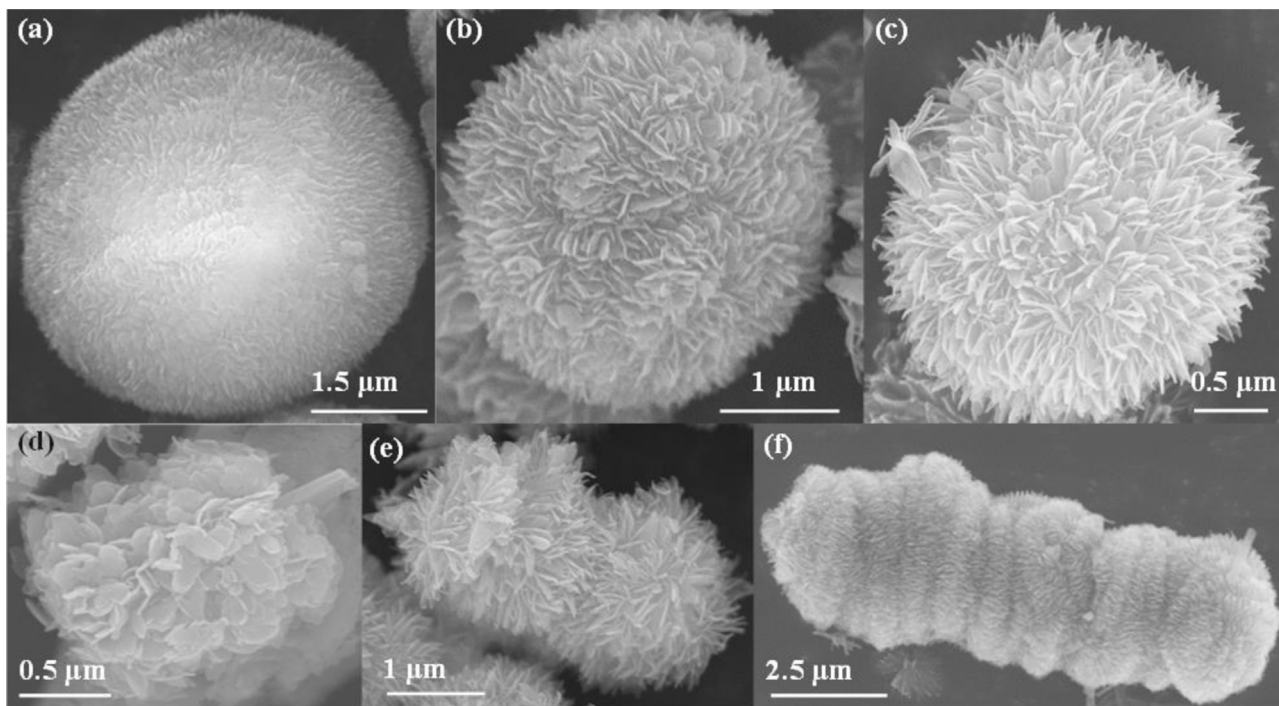
To further elucidate the microstructural features of the S-TiO<sub>2</sub> and BS-TiO<sub>2</sub> samples, transmission electron microscopy (TEM) were employed. Fig. 3a–b present the views of S-TiO<sub>2</sub> obtained from 100 C<sub>2</sub>H<sub>4</sub>O<sub>2</sub>:1.15 TBOT: 0.12 AgNO<sub>3</sub>. It is clear that S-TiO<sub>2</sub> is composite of ultra-thin TiO<sub>2</sub> nanosheets separated by nanoscale spaces. Fig. 3c shows the typical TEM image of the BS-TiO<sub>2</sub>. The amplified TEM image (Fig. 3d) of BS-TiO<sub>2</sub> clearly reveals the growth of the spheres on the TiO<sub>2</sub> nanobelt.

The unique structure features of S-TiO<sub>2</sub> and BS-TiO<sub>2</sub> provide them large surface area. N<sub>2</sub> adsorption–desorption measurements evidence that the specific surface area (S<sub>BET</sub>) of S-TiO<sub>2</sub> is around 172.7 m<sup>2</sup>/g, which is two times larger than of standard P25 nanoparticles (59 m<sup>2</sup>/g). BS-TiO<sub>2</sub> possesses an impressively high S<sub>BET</sub> of 223.8 m<sup>2</sup>/g (N<sub>2</sub> adsorption and desorption isotherms is shown in Fig. 4a), which is much higher than that of the reported quasi-1D hierarchical structures and 3D TiO<sub>2</sub> spheres (typically less than 100 m<sup>2</sup>/g) [4,5,10,17,18], indicating the potential advantage of BS-TiO<sub>2</sub> for dye-adsorption in the photoanodes of the DSCs. The crystallinity and the phase property of the as-prepared BS-TiO<sub>2</sub> samples also benefit for the use in the photoanodes of DSCs, as revealed by the X-ray diffraction (XRD) patterns of the primary TiO<sub>2</sub> nanobelt and BS-TiO<sub>2</sub> shown in Fig. 4b. TiO<sub>2</sub> nanobelt shows the combination diffraction peaks of anatase phase and TiO<sub>2</sub>-B phase. BS-TiO<sub>2</sub> after calcination at 600 °C shows the typical anatase phase, demonstrating the surface building blocks on the nanobelt, i.e. the 3D spheres, possess anatase phase. The anatase type of the 3D spheres is benefit for the use as the dye-loading matrix and electron acceptor in the photoanodes of DSCs [19].

Therefore, BS-TiO<sub>2</sub> with direct electron transport path and large specific surface area is selected to construct photoanodes in the DSCs. BS-TiO<sub>2</sub> and the standard P25-TiO<sub>2</sub> photoanodes were prepared by doctor-blading of BS-TiO<sub>2</sub> and P25-TiO<sub>2</sub> aqueous slurries, respectively, followed by calcination at 500 °C. Optical transmission spectra of BS-TiO<sub>2</sub> and P25-TiO<sub>2</sub> photoanodes are shown in Fig. 5a. The measurement was carried out using a spectrometer with an integrating sphere, taking into account the diffused light. BS-TiO<sub>2</sub> photoanode yields low transmittance, especially at the wavelength of 500–800 nm. Due to the large bandgap of TiO<sub>2</sub> (3.0–3.2 eV) [20], which only absorbs photons at the wavelength shorter than 410 nm, the reduced transmittance in BS-TiO<sub>2</sub> photoanode at long wavelengths can be ascribed to the effect of light



**Fig. 1.** Schematic illustration for the preparation of spherical TiO<sub>2</sub> (S-TiO<sub>2</sub>) and bunchy spherical TiO<sub>2</sub> hierarchical structures (BS-TiO<sub>2</sub>) from TBOT solutions with C<sub>2</sub>H<sub>4</sub>O<sub>2</sub> as the solvent.



**Fig. 2.** (a–c) Scanning electron microscopy (SEM) images of S-TiO<sub>2</sub> with different surface morphologies obtained from TBOT solutions with C<sub>2</sub>H<sub>4</sub>O<sub>2</sub> as the solvent having the composition of (a) 100 C<sub>2</sub>H<sub>4</sub>O<sub>2</sub>:2.3 TBOT: 0.95 AgNO<sub>3</sub>, (b) 100 C<sub>2</sub>H<sub>4</sub>O<sub>2</sub>:1.15 TBOT: 0.24 AgNO<sub>3</sub>, and (c) 100 C<sub>2</sub>H<sub>4</sub>O<sub>2</sub>:1.15 TBOT: 0.12 AgNO<sub>3</sub>; (d–f) SEM images of BS-TiO<sub>2</sub> obtained from TBOT solutions having the composition of 100 C<sub>2</sub>H<sub>4</sub>O<sub>2</sub>:1.15 TBOT: 0.12 AgNO<sub>3</sub> and 0.48 TiO<sub>2</sub> nanobelts with different lengths.

scattering inner BS-TiO<sub>2</sub> photoanode, as in accordance with the deduction from the morphology of BS-TiO<sub>2</sub>. Considering the absorption spectrum of N719 dye (Fig. 5a), which has one main peak located at 510 nm, the light scattering effect in BS-TiO<sub>2</sub> photoanode will favor the light absorption of N719 in the photoanode. In addition, the amount of the adsorbed dye molecules is also increased due to the large specific surface area of BS-TiO<sub>2</sub>. The dye-loading amount from BS-TiO<sub>2</sub> photoanode calculated by dye desorption is found to be 0.21 μmol/cm<sup>2</sup>, which is much higher than that of P25-TiO<sub>2</sub> photoanode, as shown in Table 1. The favored light absorption and increased dye-loading will lead to increased light harvesting in the DSC with BS-TiO<sub>2</sub> photoanode.

Photocurrent–voltage (*J*–*V*) characteristics of the DSCs based on BS-TiO<sub>2</sub> and P25-TiO<sub>2</sub> photoanodes are shown in Fig. 5b. The photovoltaic parameters extracted from *J*–*V* curves are shown in Table 1. The DSC based on BS-TiO<sub>2</sub> photoanode shows the highest short-circuit photocurrent density (*J*<sub>sc</sub>) of 14.3 mA/cm<sup>2</sup> and thus the highest power conversion efficiency (PCE) of 7.5%, which is 34% higher than these of DSC based on P25-TiO<sub>2</sub> photoanode (5.6%). Moreover, the DSC based on the bilayer photoanode including a P25-TiO<sub>2</sub> layer and a scattering layer composed by TiO<sub>2</sub> nanoparticles with the average diameter of 200 nm was also fabricated, and its photovoltaic parameters are shown in Fig. 5b and Table 1. The addition of the scattering layer to P25-TiO<sub>2</sub> photoanode has increased *J*<sub>sc</sub> of the DSC to 13.9 mA/cm<sup>2</sup> and PCE to 7.1%, as in accordance with the enhanced light harvesting by the scattering layer, whereas these values are still lower than those obtained in the DSC based on a single layer of BS-TiO<sub>2</sub> photoanode. Hence, BS-TiO<sub>2</sub> possesses great potential in the application in the photoanode of the DSCs due to their optical and electron transport advantages.

To get an insight in the carrier transport property of the photoanodes, electrochemical impedance spectroscopy (EIS) is a powerful tool for interpreting the internal resistances and the electron transport phenomena within an electrochemical system [21,22]. EIS spectra of the DSCs with different photoanodes were

carried out at the frequency range of 0.1 Hz to 100 kHz under one-sun illumination at *V*<sub>OC</sub>. Fig. 6 shows Nyquist plots obtained from the DSCs based on BS-TiO<sub>2</sub> and P25-TiO<sub>2</sub> photoanodes. The spectra show three distinguishable arcs, which are associated with the charge transfer resistance (*R*<sub>ct1</sub>) at Pt counter electrode, the recombination impedance at the TiO<sub>2</sub> photoanode (*R*<sub>ct2</sub>) and the Warburg diffusion process of redox couple (*Z*<sub>N</sub>) from the left to the right circle, respectively. The electron transport in the photoanode can be judged by the recombination impedance at the TiO<sub>2</sub> photoanode, which is estimated from the diameter of the middle semicircle by *Zview* software and listed in Table 2. The impedance is obviously small in BS-TiO<sub>2</sub> photoanode, which is agree with the speculation from its direct electron transport path. Accordingly, in the photoanode without direct electron transport path constructed by P25-TiO<sub>2</sub>, the recombination impedance is larger. The different recombination impedances of the photoanodes are determined by the recombination dynamics in the photoanodes, which can be investigated by intensity-modulated photocurrent spectroscopy (IMPS) and intensity-modulated photovoltage spectroscopy (IMVS) measurements.

IMPS and IMVS measurements were performed to analyze the electron transport and recombination dynamics of the photoanodes. The electron transport time (*τ*<sub>tr</sub>), the electron diffusion coefficient (*D*<sub>n</sub>) and the electron recombination lifetime (*τ*<sub>r</sub>) are determined from IMPS and IMVS measurements using the following schemes:

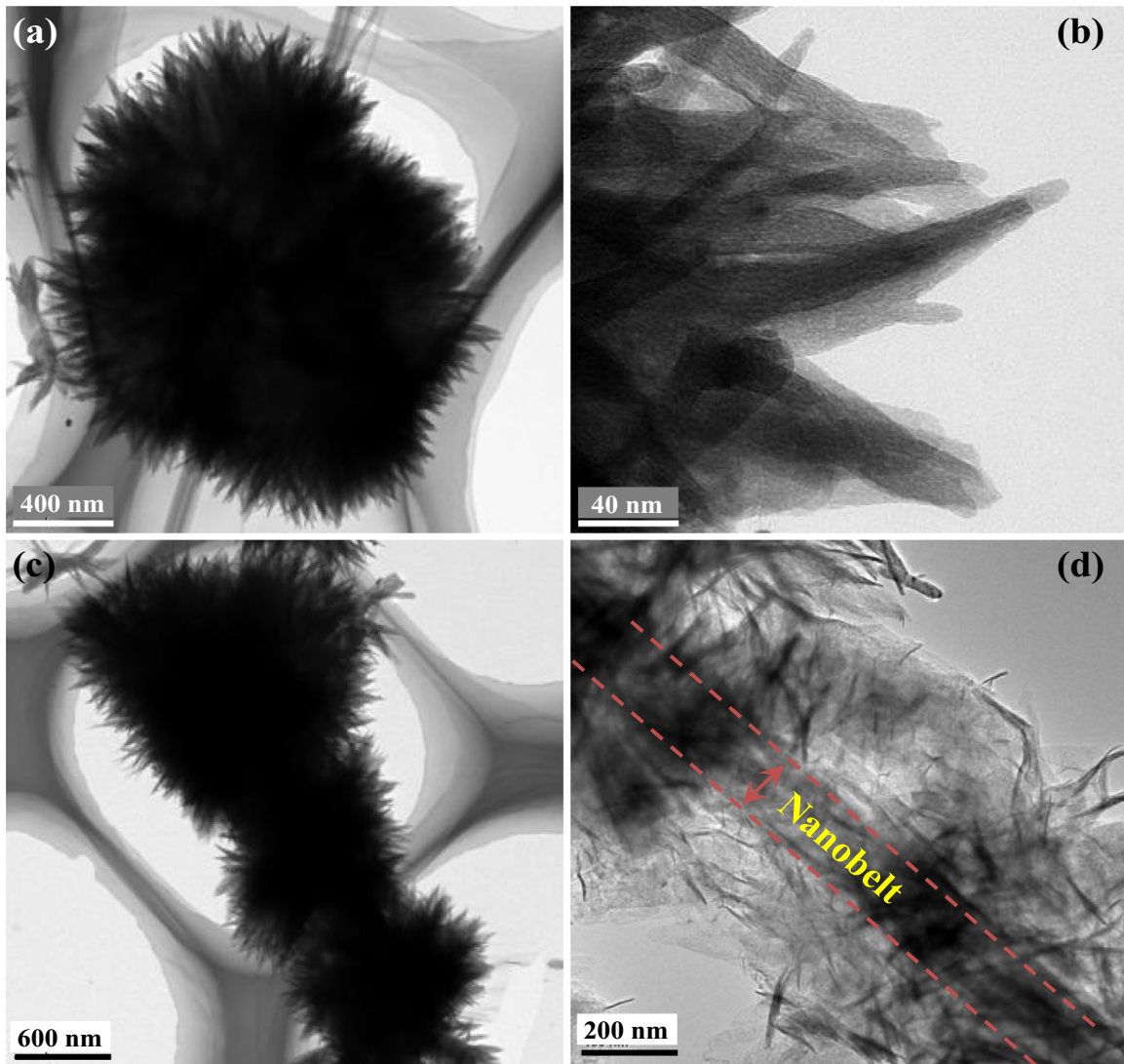
$$\tau_{tr} = 1/2\pi f_{min,IMPS} \quad (1)$$

$$D_n = d^2/C\tau_{tr} \quad (2)$$

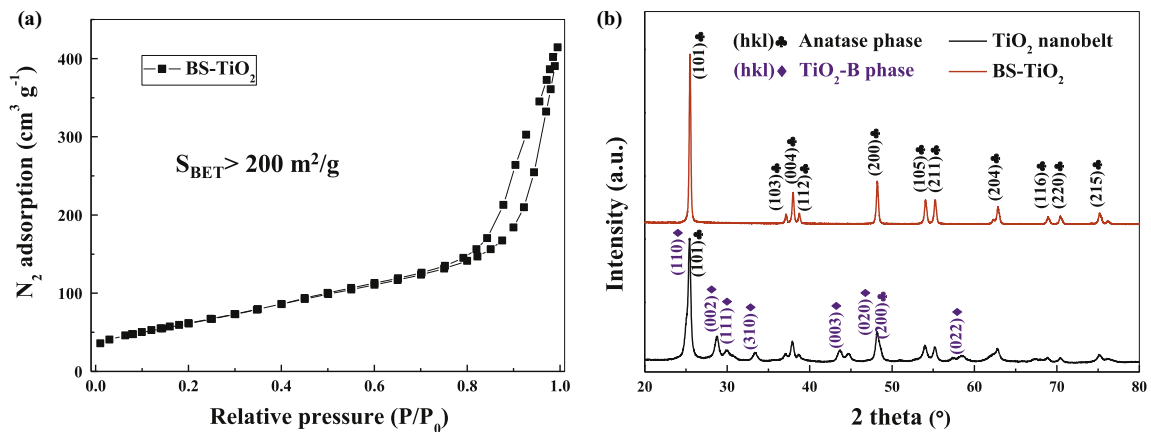
$$\tau_r = 1/2\pi f_{min,IMVS} \quad (3)$$

Where *f*<sub>min,IMPS</sub> and *f*<sub>min,IMVS</sub> represent for the frequency minimum of imaginary part in IMPS and IMVS measurements, respectively; *d*





**Fig. 3.** TEM images of S-TiO<sub>2</sub> (a,b) and BS-TiO<sub>2</sub> (c,d) samples obtained from TBOT solutions with C<sub>2</sub>H<sub>4</sub>O<sub>2</sub> as the solvent having the composition of 100 C<sub>2</sub>H<sub>4</sub>O<sub>2</sub>:1.15 TBOT: 0.12 AgNO<sub>3</sub>.



**Fig. 4.** (a) Nitrogen adsorption and desorption isotherms of BS-TiO<sub>2</sub> after calcinated at 350 °C; (b) XRD patterns of the TiO<sub>2</sub> nanobelt and BS-TiO<sub>2</sub>.

represents for the thickness of the photoanode;  $C$  is a constant which equals to 2.5. Fig. 7 shows the  $D_n$  and  $\tau_r$  values obtained from the IMPS and IMVS measurements, respectively, plotted as a function of the incident photon intensity. The highest electron diffusion coefficient was obtained from the BS-TiO<sub>2</sub> based cell,

indicating that BS-TiO<sub>2</sub> photoanode is more efficient in transporting electrons than P25-TiO<sub>2</sub> photoanode. This can be ascribed to the direct electron transport path through the 1D nanobelt trunk. The TiO<sub>2</sub> nanobelt branch in BS-TiO<sub>2</sub> is efficient to shorten the electron transfer path and can easily collect electrons from

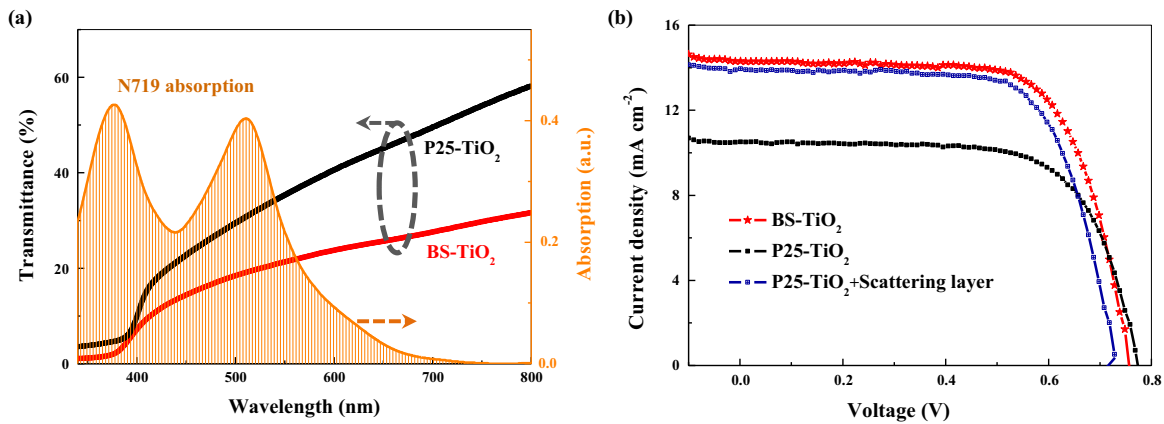


Fig. 5. (a) Transmittance spectra of the photoanodes fabricated with BS-TiO<sub>2</sub> and P25-TiO<sub>2</sub>. (b) Photocurrent–voltage ( $J$ – $V$ ) curves of the DSCs using BS-TiO<sub>2</sub> and P25-TiO<sub>2</sub> photoanodes.

Table 1

Dye-loading amount of BS-TiO<sub>2</sub> and P25-TiO<sub>2</sub> samples and the corresponding photovoltaic parameters ( $J_{SC}$ ,  $V_{OC}$ , FF, PCE) of the DSCs.

Samples	Dye-load- ing ( $\mu\text{mol}/\text{cm}^2$ )	$V_{OC}$ (V)	$J_{SC}$ ( $\text{mA cm}^{-2}$ )	FF (%)	PCE (%)
BS-TiO <sub>2</sub>	0.21	0.76	14.3	69	7.5
P25-TiO <sub>2</sub>	0.08	0.77	10.5	69	5.6
P25-TiO <sub>2</sub> +scattering layer	/	0.73	13.9	70	7.1

surface nanosheets, which improve the charge collection efficiency. On the other hand, the electron recombination lifetime of BS-TiO<sub>2</sub> photoanode is longer than that of P25-TiO<sub>2</sub> photoanode, despite of the much larger surface area of BS-TiO<sub>2</sub> which provides more trapping sites and leads to a larger charge recombination with  $I_3^-$  in the electrolyte. This can be attributed to the large grain size of BS-TiO<sub>2</sub> to establish efficient built-in potential for directing electron transport and reducing the surface recombination in BS-TiO<sub>2</sub> photoanodes.

The effective electron diffusion length ( $L_n$ ) was obtained from  $L_n = \sqrt{D_n \tau_r}$ , which is the average distance an injected electron can travel through the photoanode prior to undergoing recombination. A long  $L_n$  is important for DSCs because only a small loss of injected charge occurs in the photoanodes with a longer  $L_n$  before it is collected. As shown in Table 2, the  $L_n$  of the BS-TiO<sub>2</sub> DSC is up to 37  $\mu\text{m}$ , which is much higher than the P25-TiO<sub>2</sub> DSC, demonstrating the critical role of the nanobelt in electron transport. Furthermore,  $L_n$  of the BS-TiO<sub>2</sub> DSC is three times long of the optimal thickness (10–12  $\mu\text{m}$ ) of the photoanode, demonstrating that the generated electrons can reach the electrode with minor losses and thus, a much thicker film can be fabricated without negatively affecting the electron collection efficiency. Hence, further enhancement in the device efficiency can be achieved by optimizing the thickness of the photoanode based on BS-TiO<sub>2</sub>.

The advantages of BS-TiO<sub>2</sub> are depicted in Fig. 8. In a DSC, as shown in Fig. 8a, the dye molecules in the photoanode absorb the incident light and TiO<sub>2</sub> in the photoanode transports the photo-generated electrons. In a typical photoanode using P25-TiO<sub>2</sub> and without scattering layer, as shown in Fig. 8b, the incident light travels across the photoanode without being scattered due to the nanoscale size of the nanoparticles. Meanwhile, the electron transport in P25-TiO<sub>2</sub> based photoanode lacks inherent directionality, and hence, the driving of electrons to a conductive contact for photocurrent extraction is not efficient. Thus, to balance

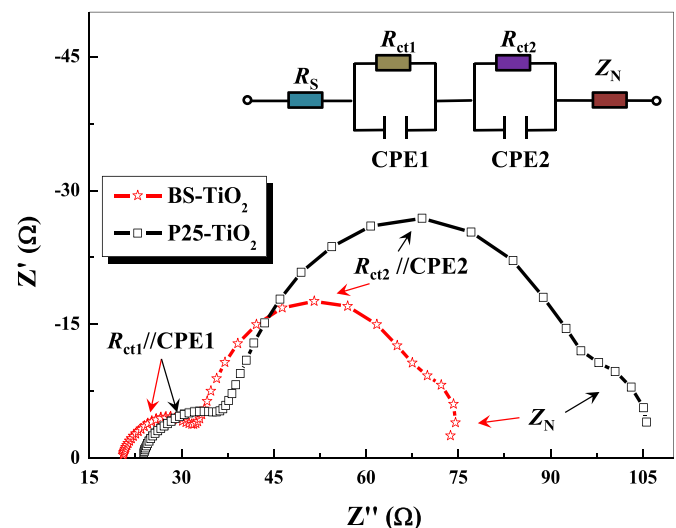


Fig. 6. EIS of DSCs based on different photoanodes. Inset shows the equivalent circuit of the DSCs. The irradiation intensity for measuring the  $J$ – $V$  curves is  $100 \text{ mW cm}^{-2}$  under AM1.5 condition.

Table 2

The kinetic parameters determined from electrochemical measurements

Samples	$R_{ct2}$ ( $\Omega$ )	$D_n$ ( $\text{cm}^2\text{s}^{-1}$ )	$\tau_r$ (s)	$L_n$ ( $\mu\text{m}$ )
BS-TiO <sub>2</sub>	40	$5.35 \times 10^{-5}$	0.25	36.8
P25-TiO <sub>2</sub>	53	$1.84 \times 10^{-5}$	0.07	11.3

the trade-off between the light harvesting and electron collection, typical P25-TiO<sub>2</sub> based photoanode yields moderate light and electron harvesting. A scattering layer is able to improve the light harvesting (Fig. 8b); however, the bilayer photoanode requires more complex fabrication processes and is not favorable for electron transport.

By replacing nanoparticulate films with quasi-1D architectures, i.e., BS-TiO<sub>2</sub> (Fig. 8c), the incident light can be scattered, which increases the optical path and thus the light harvesting efficiency. In addition, the enhanced light harvesting is also favored by the large surface area of BS-TiO<sub>2</sub> based photoanode for dye adsorption. Hence, single BS-TiO<sub>2</sub> layer is more efficient than P25-TiO<sub>2</sub>/scattering layer in light harvesting. Furthermore, the photogenerated electrons are

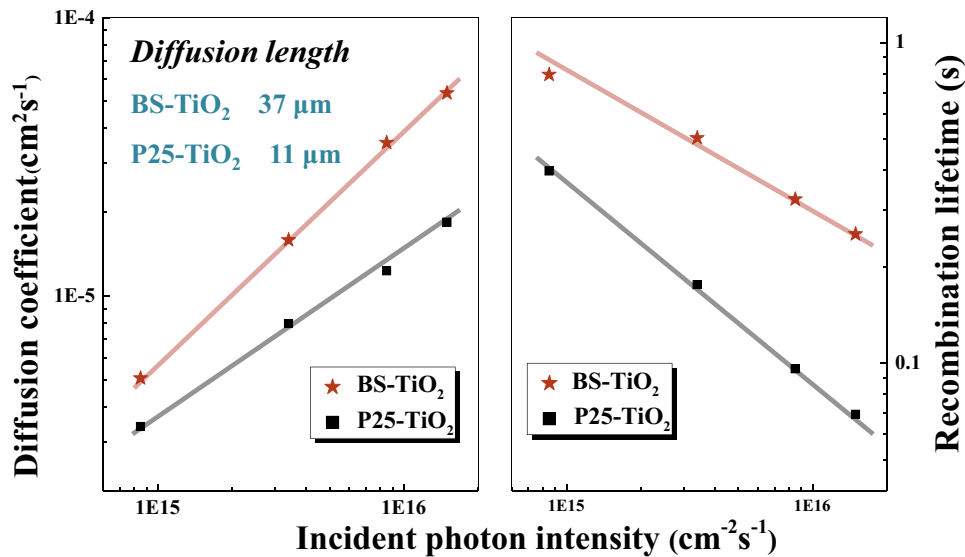


Fig. 7. Diffusion coefficient ( $D_n$ ) and electron recombination lifetime ( $\tau_r$ ) of DSCs fabricated with different photoanodes, as determined from IMPS/IMVS measurements.

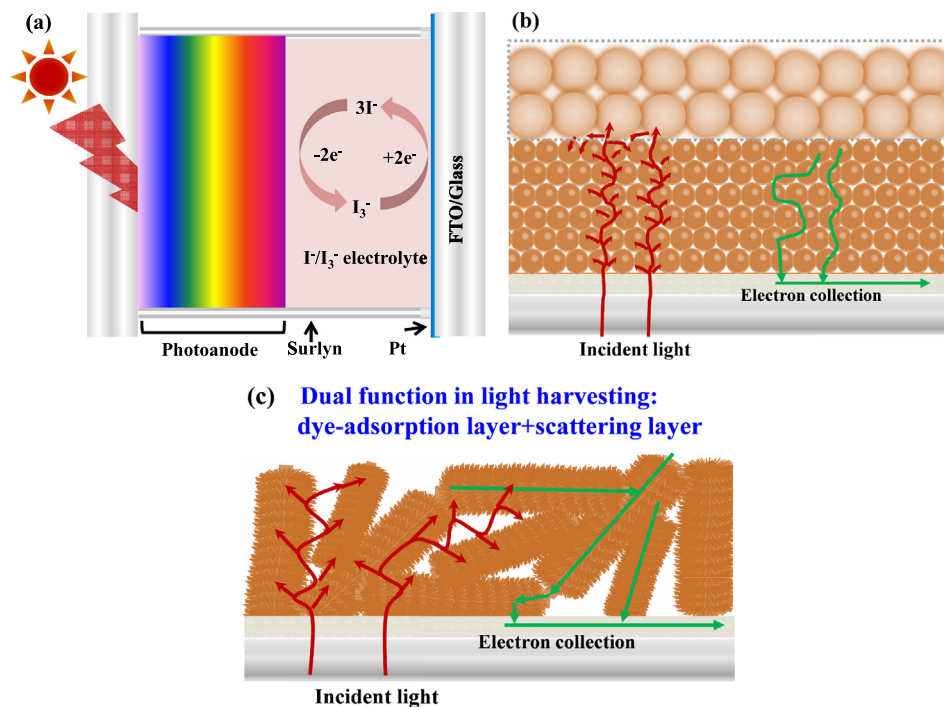


Fig. 8. (a) Schematic structure of a DSC; (b, c) Schematic mechanisms of the light transport and electron collection processes in different photoanodes, (b) P25-TiO<sub>2</sub> with and without scattering layer, and (c) BS-TiO<sub>2</sub>.

allowed to move only in one dimension instead of randomly in three, leading to the much long diffusion length in BS-TiO<sub>2</sub> based photoanode. Hence, BS-TiO<sub>2</sub> photoanode can effectively improve the light harvesting and electron collection, leading to the enhanced photovoltaic performance of DSCs.

Hence, the excellent performance of the DSC using BS-TiO<sub>2</sub> photoanode is due to the increased light harvesting and electron collection through its large surface area for dye-loading, improved light scattering ability with their wavelength scale size and fast electron transport path through the 1D nanobelt. Furthermore, the unique properties of BS-TiO<sub>2</sub> with large specific surface area and fast electron transport also imply its great potential for the application in many fields including perovskite solar cells, photocatalytic and other optoelectronic devices.

### 3. Experimental

#### 3.1. Fabrication of bunched TiO<sub>2</sub> hierarchical structures

TiO<sub>2</sub> spheres were fabricated through hydrothermal synthesis with TBOT and acetic acid as the reactants and AgNO<sub>3</sub> as the agent [23]. Bunched TiO<sub>2</sub> hierarchical spheres were obtained by introducing TiO<sub>2</sub> nanobelt [24] as the trunk in the synthesis of TiO<sub>2</sub> spheres. The concentration ratio of the reactants and AgNO<sub>3</sub> was varied to control the surface morphology of the TiO<sub>2</sub> structures.

#### 3.2. Fabrication of DSCs.

TiO<sub>2</sub> photoanodes were prepared by doctor-blading of TiO<sub>2</sub> slurries on FTO/glass substrates. All of the TiO<sub>2</sub> photoanodes were

post-treated with  $\text{TiCl}_4$ . After calcination,  $\text{TiO}_2$  photoanodes were immersed in 0.3 mmol/l ethanol solution of N719 dye for 24 h. DSCs were fabricated by assembling dye-sensitized  $\text{TiO}_2$  photoanodes with sputtered Pt CEs [25] using 30  $\mu\text{m}$  thick Surlyn.  $\text{I}^-/\text{I}_3^-$  electrolyte with acetonitrile as the solvent was used. Symmetric cells for electrochemical measurements were fabricated by assembling two identical CEs together using 30  $\mu\text{m}$  thick Surlyn.

### 3.3. Characterization of the materials and DSCs

The structure and morphology of the materials were characterized by transmission electron microscopy (TEM, FEI Tecnai F20) and scanning electron microscopy (SEM, FEI Quanta 200F). Current density–voltage curves of DSCs were measured using a source meter (Keithley 2400) under AM 1.5 G irradiation with a power density of 100  $\text{mW}/\text{cm}^2$  from a solar simulator (XES-301S+EL-100). Electrochemical measurements were carried out using the electrochemical workstation (Zahner zennium) performed on DSCs under irradiation from solar simulator for EIS measurement and from light emitting diode (535 nm) for IMVS/IMPS measurements, respectively.

## 4. Conclusion

In conclusion, bunched  $\text{TiO}_2$  hierarchical spheres, consisting of 1D  $\text{TiO}_2$  nanobelt and multiple  $\text{TiO}_2$  microspheres constructed by nanosheets, were rationally designed and fabricated to enhance the light harvesting and electron collection in photoanodes of dye-sensitized solar cells. These bunched  $\text{TiO}_2$  hierarchical spheres exhibit faster electron transport properties and long electron diffusion length compared to  $\text{TiO}_2$  nanoparticles. Moreover, they also show large specific surface area and improved light scattering. The excellent electron collection and light harvesting abilities of bunched  $\text{TiO}_2$  hierarchical spheres enable the enhancement of the conversion efficiency of the related solar cells. This work highlights the rational design of the  $\text{TiO}_2$  nanostructures for highly efficient dye-sensitized solar cells, which also implies the application of the bunched  $\text{TiO}_2$  hierarchical spheres in other optoelectronic fields.

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